

March 2012

**Bond resonance in superconducting rapid cooled alloys:
 $(Bi_{1.7}Pb_{0.3}Sr_2Can-1CunO_{2n+4+\delta})_2$, n=1 to 9 detected by novel
local atomic enhanced XRD**

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Recommended Citation

K Tyson, J Kmiec, Juana Acrivos, D. Gulamova, and J Chigvinadze. "Bond resonance in superconducting rapid cooled alloys: $(Bi_{1.7}Pb_{0.3}Sr_2Can-1CunO_{2n+4+\delta})_2$, n=1 to 9 detected by novel local atomic enhanced XRD" *National ACS Meeting* (2012).

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Bond resonance in superconducting rapid cooled alloys: $(Bi_{1-x}Pb_0.3Sr_2Ca_{n-1}Cu_xO_{2n+4+\delta})_2$, $n=1$ to 9 detected by novel local atomic enhanced XRD

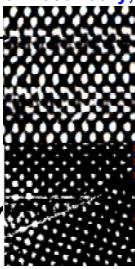
National ACS Meeting
San Diego CA, March 2012

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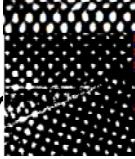
Abstract: We discuss bond resonance in high temperature layer superconducting cuprates, HTLSC alloys: $(Bi_{1-x}Pb_0.3Sr_2Ca_{n-1}Cu_xO_{2n+4+\delta})_2$, $n=1$ to 9 called $(2,2:2:n-1:n)$ prepared in Uzbekistan from melts in concentrated sun flux, followed by rapid cooling, SFQA technology that preserves melt tiling, $T_{c,n}$ measured in Tbilisi, and local atomic enhanced synchrotron X-ray diffraction, XRD near the Cu K-edge at DOE National Laboratory SLAC-SSRL. The SFQA alloys structure indexed in the ideal D^{17}_{4h} Space Group indicates $(n-1)$ units: $(CuO_{2/1}/Ca/(CuO_{2/1})_2$ are intercalated at each end of the $n=1$ phase, $[(CuO_{2/1}/SrO/Bi_{1-x}Pb_0.3O/Bi_{1-x}Pb_0.3O/CuO_{2/1})_2]$. Surface effects lead to different n -phase mixing, and periodic lattice distortions, PLD, through bond resonance in crystal axes: $2(r_{Cu^{2+}}+r_{O^{2-}})=4.2\text{\AA}=a=3.815\text{\AA}>(CuO_{\text{organic}}+(O=O-O)/2^{3/2})=3.2\text{\AA}$, $c=26.4\text{\AA}$, $c_n-c_1=1.64(n-1)a$, or $\delta c_n/(2a(n-1))\approx 0.2$. Local atomic bonds Cu-Ca, Cu-Sr, Cu-O are detected by local atomic inelastic scattering, $T_{c,n}$ related to bond resonance in the family predicts $T_{c,n}\approx 298K$ as $n\approx 50$, if the superconducting skin depth is within a few μm .

Introduction: Covalent vs ionic materials with grain boundary, GB:

Covalent vs Ionic
1. HRTEM 2212: 1998
WY LIANG et al.



2. Non-bonding
 Cu_4O_4 HOMO: 1993



JV Acrivos,
O Stradella
3.5SrTiO3 Σ 13:[015]GB
2001, K vanBenthem

Figure 1: 2212, and ionic $SrTiO_3$ electron density, by high resolution electron microscopy, HRTEM^[1]

Cartoon of bc plane front:■, $a/2$ back:■ ID:
Bi, Sr, Cu, Ca, O, vacant
 $SrTiO_3[01L]$ GB are also present in 2:2:n-1:n [1].

RESULTS:

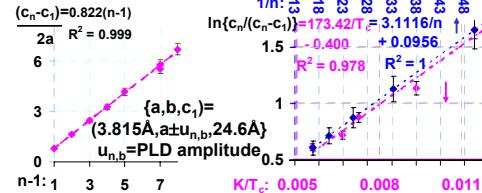


Figure 2: Activity $\ln[c_n/(c_n - c_1)]$ related to $T_{c,n}$ data obtains:
 $-\Delta G^*_n = k_B T \ln[c_n/(c_n - c_1)] = k_B T [3.112/n + 0.096]$

$$-\Delta G^*_n/k_B T = 173.42/T_{c,n} - 0.400.$$

Discussion: 1. Resonance is indicated by crystal axes: $4.2\text{\AA}=2(r_{Cu^{2+}}+r_{O^{2-}})>a=3.82\text{\AA}>(CuO_{\text{organic}}+(O=O-O)/2^{3/2})=3.2\text{\AA}$ is an average of purely ionic and covalent bonds. 2. The CuO non bonding Cu_4O_4 HOMO allows [10L] GB formation in rapid cooled sun flux melts. These induce rotation detected by local Cu atomic enhanced XRD of pairs Q_0 , Q^* of mixed $n=4$ and $n=8$ phases formed in melt, indicating strong local Cu-Ca, Cu-Sr, Cu-O bonds. 3. Interaction between extended electronic states $|Q_0\rangle$, $|Q^*\rangle$ decrease Gibbs free enthalpy of formation, ΔG^* related to $T_{c,n}$ (fig. 3) that may be due to tiling in melts exposed to sun radiation. **Conclusion:** HTLSC, alloys grown by SFQA novel fast cooling, which preserves melt tiling, mix different n phases, and show GB typical of parent $SrTiO_3$ promise useful applications with increased $T_{c,n} \approx 150$ to 180K in emerging isolated regions, which appear to be related to Gibbs free enthalpy of formation ΔG^*_n by tiling.

ACKNOWLEDGEMENTS: Support for truly interdisciplinary, international work at San José State and DOE National Laboratory SLAC-SSRL, USA; Tbilisi, Georgia and Tashkent, Uzbekistan by the Science and Technology Centre in Ukraine (STCU) #4266 and Georgian National Science Foundation (GNSF) grant 08-477-4-260 and the Camille and Henry Dreyfus Foundation and DOE, and for published and unpublished HRTEM by YW Liang^[1] is gratefully acknowledged.

References: ^[1] (a) W. Zhou, D-A. Jefferson and W.Y. Liang, "Research Review, High Temperature Superconductivity" Cambridge University, WY Liang, editor (1998), p.15; (b) W.Y. Liang and Y. Yin unpublished (1998); (c) K van Benthem, R.H French, W Sigle, C Eisasser, M. Rühle, Ultramicroscopy, 86, 303, (2001); (d) J.V. Acrivos, Physics and Chemistry of Electrons and Ions in Condensed Matter, J.V. Acrivos, N.F. Mott and D.A. Yoffe, ed., NATO ASI Series C130, 479 (1983); ^[2] J.G Chigvinadze, AA lashvili, TV Machaidze, Phys. Letts. A 300, 524 (2002); ^[3] (a) DD Gulamova, DE Uskenbaev, G Fantozi, JG Chigvinadze, O.V. Magradze, Zh. Tekhn. Fiz., 79, 98 (2009); (b) T Scopigno et al., Physica B 318, 341 (2002); (c) J.V. Acrivos, Microchemical Journal 99,239-245 (2011)

K_0 incident XRD, scattered to K_1 at $h\nu$ resonance, near the Cu K-edge, $Q=K_1-K_0: 0|a_0a_0H_{\text{OMO}}>\otimes_a(\exp(iK_0\cdot R\cdot \mathbf{m}))\cdot h\nu>^a(a-1)s_a a_0H_{\text{OMO}}(Q+d)_{\text{LUMO}}>\otimes_a\exp(iK_0\cdot R\cdot \mathbf{m})$ followed in $t\leq 10^{-17}\text{s}$ by core state depletion of elements, a_0 bonded to excited a , by emission/absorption/ backscattering from Q_{HOMO} and $(Q+d)_{\text{LUMO}}$ with ϵ_0 , ϵ_1 in the Q plane, add ΔE to $h\nu$, determined by atomic f_{Cu} , n neighbor χ_{Cu} tensors. E levels of possible atomic excitations after t_0 , E_0 near Cu K-edge in HTLSC

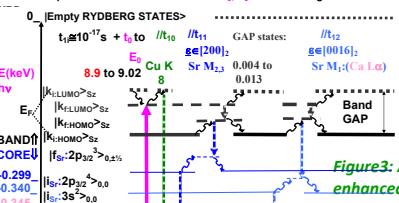


Figure 3: Atomic Cu enhanced XRD in planes Q_0 , Q^* obtained $Q_{\text{RMS}}=Q^*+\sum Q_i^*$, $\frac{1}{2}Q=2(Q-Q_0)$ if $iQ_0=Q^*$ in proximity.

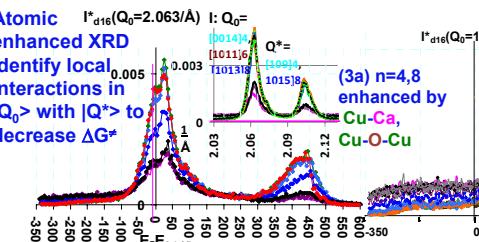


Figure 4: Atomic enhanced XRD plots for n=4, n=8, and n=7.

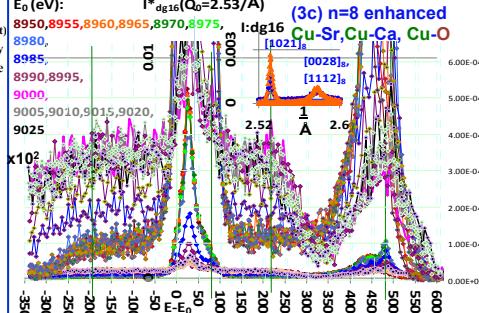


Figure 5: XRD patterns for various Cu K-edge positions.

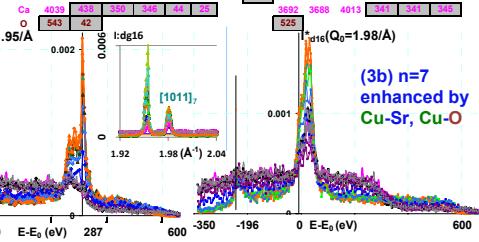


Figure 6: XRD patterns for various Cu K-edge positions.

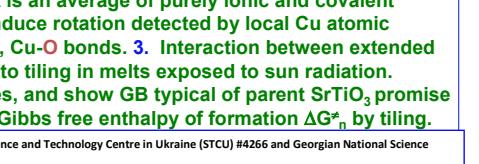


Figure 7: XRD patterns for various Cu K-edge positions.